

## Co-ordination Polymers of Rhodium(1+) with Di-isocyanonaphthalene (1,4 and 1,5) Linkages

By AVI EFRAITY,\* IRENE FEINSTEIN, FELIX FROLOW, and ALEXANDER GOLDMAN

(Departments of Organic Chemistry and Structural Chemistry, The Weizmann Institute of Science, Rehovot, Israel)

**Summary** The template polymerization of tetragonal rhodium(1+) with 1,4- and 1,5-di-isocyanonaphthalenes gives two-dimensional co-ordination polymers with a stacked layers arrangement, the powder X-ray diffractometric traces of these polymers are presented and discussed

THE isomers 1,4- and 1,5-di-isocyanonaphthalene<sup>1</sup> can be regarded as non-chelating bidentate ligands capable of forming stereochemically rigid linkages between metal nuclei with a bridge span of *ca* 12 Å. Viewing the terminal isocyanide-to-metal bonds as vectors suggests that the terms 'colinear' (1,4) and 'parallel' (1,5) may be used to describe the bridging geometries of these ligands. Compared with the colinear (1,4) ligand, a displacement of 2.42 Å between the parallel vectors in 1,5-di-isocyano-

naphthalene requires a slightly longer bridge span (*ca* 0.2 Å). The co-ordination properties of these bridging ligands were ascertained for a tetragonal rhodium(1+) system.

Di-isocyanonaphthalenes (1,4 and 1,5) react with [Rh(CO)<sub>2</sub>Cl]<sub>2</sub> in CH<sub>2</sub>Cl<sub>2</sub> solution to afford quantitative yields of insoluble green polycrystalline polymers which analyse as [Rh(bridge)<sub>2</sub><sup>+</sup>Cl<sup>-</sup>]<sub>n</sub>. These polymers readily absorb water, and therefore, unless rigorously dried *in vacuo*, are best formulated as [Rh(bridge)<sub>2</sub><sup>+</sup>Cl<sup>-</sup>·xH<sub>2</sub>O]<sub>n</sub> (x = 0-5). A single strong ν(NC) at 2040 ± 5 cm<sup>-1</sup> in their IR spectra may suggest the presence of terminally co-ordinated isocyanides in a symmetric environment.

Two-dimensional models of the [Rh(bridge)<sub>2</sub>]<sub>n</sub> network which depict the template polymerization of tetragonal Rh<sup>I</sup> with 1,4-di-isocyanonaphthalene [Figure (A)] and

TABLE. Powder X-ray diffraction angles ( $\theta$ ), interplanar spacings ( $d$ ),  $\sin^2\theta$  ratio ( $N$ ), and assigned indices for the new polymers (A) and (B).

(A) <sup>c</sup>			(B) <sup>d</sup>			Assigned indices		
$\theta/^\circ$	$d/\text{\AA}$	$N_{\text{exp.}}^a$ [ $N_{\text{theor.}}^b$ ] <sup>b</sup>	$\theta/^\circ$	$d/\text{\AA}$	$N_{\text{exp.}}^a$ [ $N_{\text{theor.}}^b$ ] <sup>b</sup>	$h$	$k$	$l$
3.95	11.18	1 [1]	3.73	11.84	1 [1]	1	0	0
5.80	7.62	2.15[2]	5.50	8.04	2.17[2]	1	1	0
7.75	5.71	3.83[4]	7.55	5.86	4.08[4]	2	0	0
8.70	5.09	4.82[5]	8.45	5.24	5.10[5]	2	1	0
12.10	3.67	9.25[9]	11.35	3.91	9.15[9]	3	0	0
13.10	3.39	(10.82) <sup>e</sup>	13.05	3.41	(12.05) <sup>e</sup>	0	0	1

<sup>a</sup>  $N_{\text{exp.}} = \sin^2\theta(hk0)/\sin^2\theta(100)$ . <sup>b</sup>  $N_{\text{theor.}} = h^2 + k^2$ . <sup>c</sup> Space group  $P4$ ; cell dimensions:  $a = 11.18$ ,  $c = 3.39$  Å. <sup>d</sup> Space group  $P4$ ; cell dimensions:  $a = 11.84$ ,  $c = 3.41$  Å. <sup>e</sup> This value does not conform with integral  $h^2 + k^2$  terms.

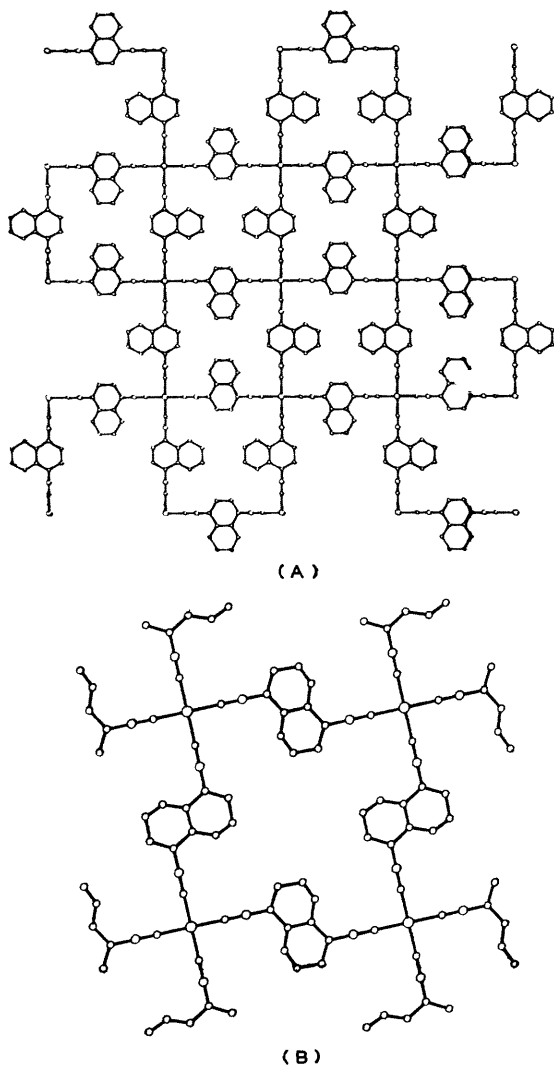


FIGURE. Two-dimensional models of the  $[\text{Rh}(\text{bridge})_2]_n$  network in the 1,4- (A) and 1,5- (B) di-isocyanonaphthalene  $\text{Rh}^{\text{I}}$  polymers.

† The cell dimension of (A) is based solely on the rhodium metal array, and does not take into account the presumed benzo ring orientations which require a cell twice its length.

<sup>1</sup> I. Ugi, 'Isonitrile Chemistry,' Academic Press, New York, 1971, and references therein.

<sup>2</sup> K. R. Mann, N. S. Lewis, R. M. Williams, H. B. Gray, and J. G. Gordon, II, *Inorg. Chem.*, 1978, **17**, 828.

<sup>3</sup> H. Endres, N. Gottstein, H. J. Keller, and R. Martin, *Z. Naturforsch., Teil B*, 1979, **34**, 827.

<sup>4</sup> H. Lipson and H. Steeple, 'Introduction to X-Ray Powder Diffraction Patterns,' Macmillan, London, 1970.

<sup>5</sup> A. Efraty, I. Feinstein, F. Frolow, and L. Wackerle, *J. Am. Chem. Soc.*, in the press.

1,5-di-isocyanonaphthalene [Figure (B)] were constructed using idealized bond angles and distances similar to those found in related systems.<sup>2,3</sup> Model calculated cell dimensions are:  $a = 11.81$  Å for (A)† and  $a = 12.04$  Å for (B). Based on these models, the shortest interbridging C---C distances in (A) (ca. 4.6 Å) and (B) (ca. 4.3 Å) suggest the absence of steric constraint. In view of the established tendency<sup>2</sup> of the  $\text{Rh}(\text{CNR})_4^+$  cation to associate *via* weak  $\text{Rh}^{\text{I}} \cdots \text{Rh}^{\text{I}}$  interactions, the three-dimensional model structures of these polymers may be derived by the eclipsed stacking of the two-dimensional models.

The powder X-ray diffractometric traces of the new polymers, measured on a Phillips Diffractometer using filtered Cu radiation and a scan speed of  $1^\circ (2\theta)$  per min, are especially informative (Table) concerning their structures. Interpretation<sup>4</sup> of the diffraction data in terms of reflections originating from Rh-containing sets of planes appears consistent with the presence of a tetragonal system in these polymers. The most intense reflection in each trace was assigned to the 100 planes, whose spacings [ $d(100)$ ] correspond to the bridge spans. Other  $hk0$  reflections were assigned according to a known technique<sup>4</sup> by considering the  $N_{\text{exp.}}$  and  $N_{\text{theor.}}$  terms (Table). One reflection in each trace, not belonging to the  $hk0$  family, was assigned to the 001 planes. This assignment suggests the intermetallic distances [ $d(001)$ ] 3.39 Å in (A) and 3.41 Å in (B), which are compatible with the distances 3.19,<sup>2</sup> 3.21,<sup>3</sup> and 3.25 Å<sup>3</sup> found in related dinuclear complexes of type  $[\text{Rh}_2(\text{CNAr})_8]^{2+}$ , as well as with distances in the range 3.21–3.36 Å characterized in similar tetragonal polymers of rhodium(1+) with other rigid di-isocyanide linkages.<sup>5</sup>

Compared with the respective model calculated terms, the experimentally determined bridge span in (A) is found to be significantly shorter (ca. 0.63 Å), whereas that in (B) is only slightly shorter (ca. 0.20 Å). These differences may be attributed, in part, to the non-idealized bond angles expected in the real polymers. The bridge span in (B) has indeed been found to be longer than that in (A), as was anticipated.

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